

# Al-Fe-Pd (Aluminum-Iron-Palladium)

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The previous review of this system by [1992Rag] presented a pseudobinary section along the Fe-AlPd join and an isothermal section at 500 °C, with the latter based on the work of [1988Fil]. No ternary compounds were reported in the Al-rich region at 500 °C. Recently, [2004Bal] investigated the phase relationships in the temperature range of 1020 to 900 °C and found the occurrence of several ternary phases in Al-rich alloys.

## Binary Systems

The Al-Fe phase diagram [1993Kat] shows that the face-centered cubic (fcc) solid solution based on Fe is restricted by a  $\gamma$  loop. The body-centered cubic (bcc) solid solution  $\alpha$  exists in the disordered A2 form, as well as the ordered B2 and D0<sub>3</sub> forms. Apart from the high-temperature phase  $\varepsilon$ , there are three other intermediate phases in the system: FeAl<sub>2</sub> (triclinic), Fe<sub>2</sub>Al<sub>5</sub> (orthorhombic), and FeAl<sub>3</sub> or Fe<sub>4</sub>Al<sub>13</sub> (monoclinic). In the Al-Pd system [1986McA], a number of intermediate phases occur: Al<sub>4</sub>Pd (hexagonal), Al<sub>3</sub>Pd<sub>2</sub> (orthorhombic); Al<sub>3</sub>Pd<sub>2</sub> (Al<sub>3</sub>Ni<sub>2</sub>-type hexagonal), three modifications of AlPd (two cubic and one rhombohedral), Al<sub>3</sub>Pd<sub>5</sub> (Ge<sub>3</sub>Rh<sub>5</sub>-type orthorhombic), AlPd<sub>2</sub> (Co<sub>2</sub>Si-type orthorhombic), and Al<sub>2</sub>Pd<sub>5</sub> (Ga<sub>2</sub>Pd<sub>5</sub>-type orthorhombic). Recently, [2001Yur] reported two more high-temperature orthorhombic phases,  $\varepsilon_6$  and  $\varepsilon_{28}$ , close to the Al<sub>3</sub>Pd composition. The Fe-Pd phase diagram [1982Kub] is characterized by the presence of a continuous solid solution  $\gamma$  in the temperature range of 1300 to 900 °C between the Fe-based fcc phase and Pd. Two superstructures FePd (AuCu type tetragonal) and FePd<sub>3</sub> (AuCu<sub>3</sub> type cubic) form

congruently from  $\gamma$  at 790 and 820 °C, respectively, and have appreciable homogeneity ranges at lower temperatures.

## Ternary Phases

[2004Bal] studied the phase equilibria in Al-rich alloys in the temperature range of 1020 to 900 °C and found three ternary cubic phases designated C (space group  $Pm\bar{3}$ ,  $a = 0.7655$  nm), C<sub>1</sub> ( $Im\bar{3}$ ,  $a = 1.5389$  nm), and C<sub>2</sub> ( $Fm\bar{3}$ ,  $a = 1.5510$  nm), each with a significant homogeneity range. An orthorhombic phase O (space group  $Cmca$ ,  $a = 1.5499$ ,  $b = 0.8102$ , and  $c = 2.3848$  nm) occurs close to the monoclinic binary phase FeAl<sub>3</sub>. In the ternary region adjacent to the binary high-temperature orthorhombic phases  $\varepsilon_6$  and  $\varepsilon_{28}$  of the Al-Pd system, two additional structure variants,  $\varepsilon_{16}$  and  $\varepsilon_{22}$ , were found. [2004Bal] clubbed all the  $\varepsilon$  variants together and designated the ternary region as  $\varepsilon$  in the isothermal section at 900 °C.

## Ternary Isothermal Sections

Using starting metals of purity 99.999% Al, 99.99% Fe, and 99.95% Pd, [2004Bal] induction-melted alloy compositions under an argon atmosphere. The samples were annealed at 1020, 995, 975, and 900 °C for 24 to 2043 h. The phase equilibria were studied with scanning and transmission electron microscopy, and the x-ray diffraction technique. The compositions of the phases were determined by the inductively coupled plasma optical emission spectroscopy and by energy-dispersive x-ray analysis. The melting

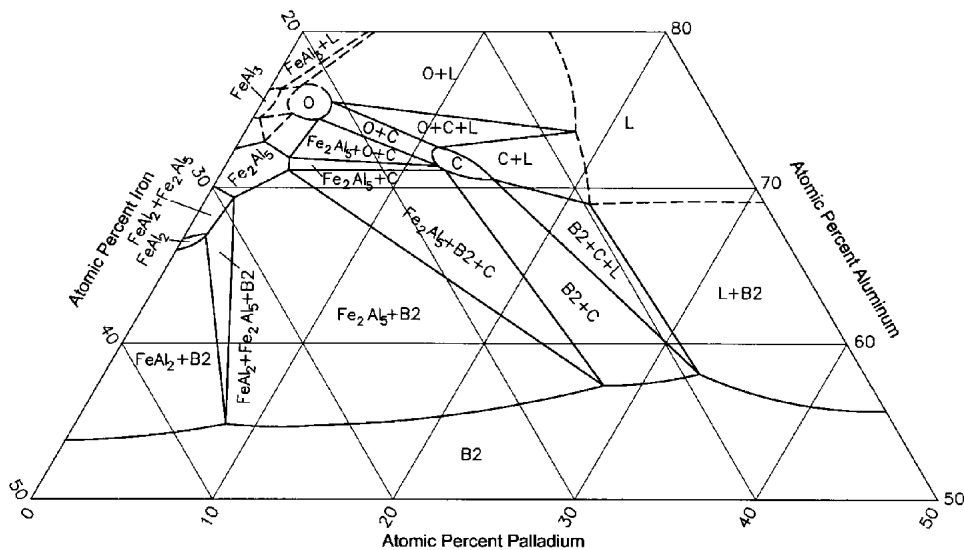


Fig. 1 Al-Fe-Pd isothermal section at 1020 °C [2004Bal]

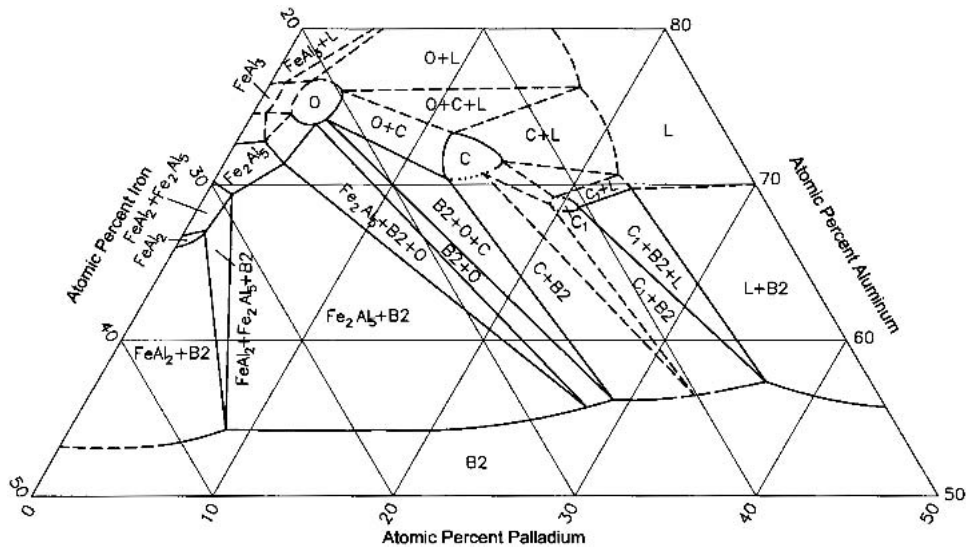


Fig. 2 Al-Fe-Pd isothermal section at 995 °C [2004Bal]

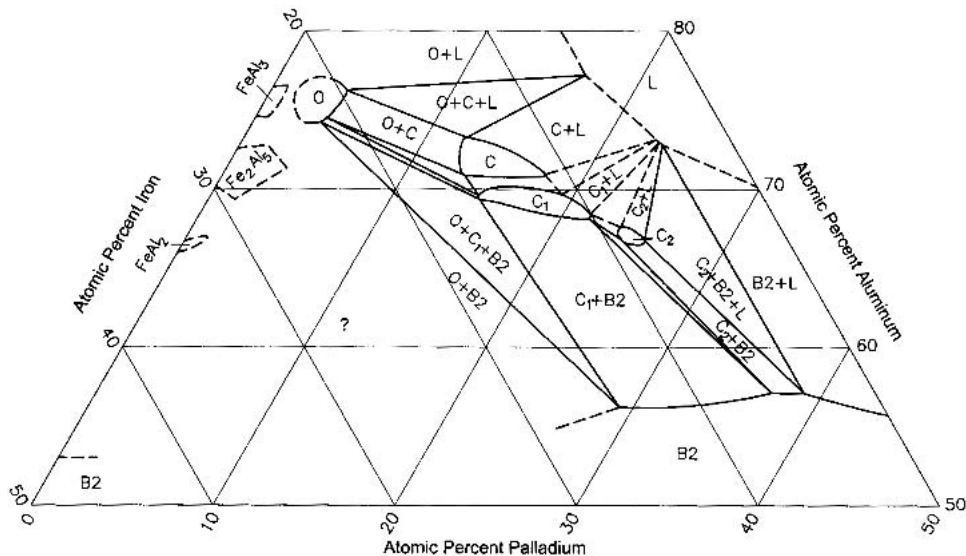


Fig. 3 Al-Fe-Pd isothermal section at 975 °C [2004Bal]

temperatures of the phases were determined by differential thermal analysis. The isothermal sections constructed by [2004Bal] at 1020, 995, 975, and 900 °C are redrawn in Fig. 1 to 4 to agree with the accepted binary data.

At 1020 °C (Fig. 1), the ternary phases *C* and *O* are stable. The *O* phase is centered around the composition (at.%)  $\text{Al}_{75.4}\text{Fe}_{21.8}\text{Pd}_{2.8}$  and dissolves up to 4.2 at.% Pd. The *C* phase has a composition range of  $\text{Al}_{72.4}\text{Fe}_{16.4}\text{Pd}_{11.2}$  to  $\text{Al}_{70.4}\text{Fe}_{14.1}\text{Pd}_{15.5}$ . The CsCl-type *B2* phase dissolves up to 57.8 at.% Al at 33.4 at.% Pd.  $\text{FeAl}_2$ ,  $\text{Fe}_2\text{Al}_5$ , and  $\text{FeAl}_3$  dissolve up to 1.4, 3.9, and 1.0 at.% Pd, respectively. At 995 °C (Fig. 2), the ternary phases *C*, *C*<sub>1</sub>, and *O* are present. The *O* phase contains ~74 to 76.5 at.% Al and up to ~4.4 at.% Pd. The *C* phase is centered around the composition  $\text{Al}_{71.6}\text{Fe}_{14.8}\text{Pd}_{13.6}$ . The homogeneity range of *C*<sub>1</sub> is small,

around  $\text{Al}_{68.5}\text{Fe}_{11.0}\text{Pd}_{20.5}$ . The solubility in the *B2* phase is up to 57.4 at.% Al at 37.0 at.% Pd.  $\text{FeAl}_2$ ,  $\text{Fe}_2\text{Al}_5$ , and  $\text{FeAl}_3$  dissolve about 1.2, 3.4, and 1 at.% Pd, respectively.

At 975 °C (Fig. 3), four ternary phases, *C*, *C*<sub>1</sub>, *C*<sub>2</sub>, and *O*, are stable. The composition range of *C* is wider than at 995 °C and extends to higher Pd contents. The homogeneity region of *C*<sub>1</sub> is also wider, compared to that at 995 °C and ranges from  $\text{Al}_{69.5}\text{Fe}_{15.7}\text{Pd}_{14.8}$  to  $\text{Al}_{68.2}\text{Fe}_{10.2}\text{Pd}_{21.6}$ . The *C*<sub>2</sub> phase occupies a narrow region around  $\text{Al}_{67.0}\text{Fe}_{8.5}\text{Pd}_{24.5}$ . At 900 °C (Fig. 4), in addition to the above four ternary phases, the  $\epsilon$  structural variants, which form below 940 °C, are present, and the region is marked as a single phase  $\epsilon$  in Fig. 4, with a range of  $\text{Al}_{75.5}\text{Fe}_{10}\text{Pd}_{14.5}$  to  $\text{Al}_{72.1}\text{Fe}_{4.8}\text{Pd}_{23.1}$  [2004Bal]. The  $\epsilon_{16}$  variant occurs at higher Fe concentrations, and the  $\epsilon_{22}$  variant occurs at lower Fe concentrations.

## Section II: Phase Diagram Evaluations

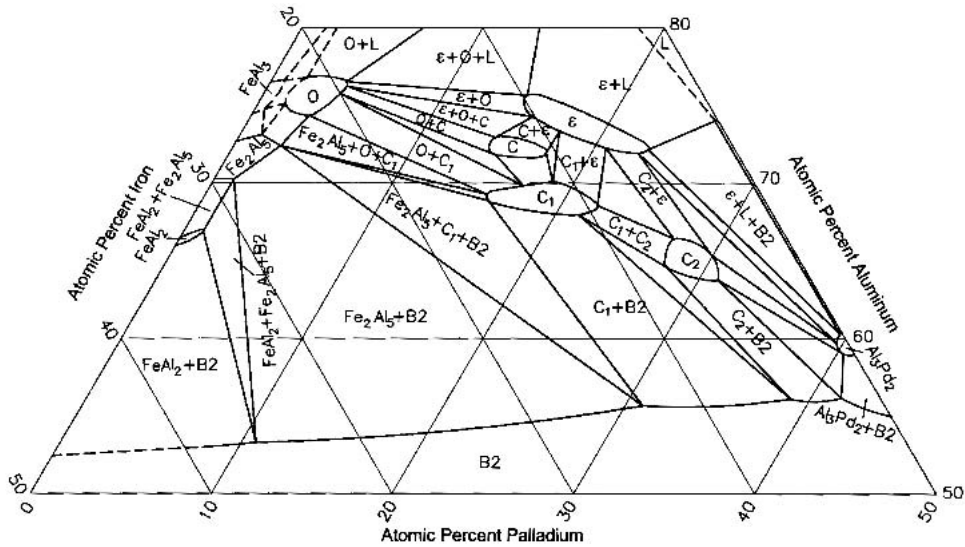


Fig. 4 Al-Fe-Pd isothermal section at 900 °C [2004Bal]

The B2 phase dissolves 56.2 at.% Al at 41.8 at.% Pd. The Pd solubility in  $\text{FeAl}_2$  is 1 at.%, and the Fe solubility in  $\text{Al}_3\text{Pd}_2$  is 1.1 at.%. The C, C<sub>1</sub>, and C<sub>2</sub> phases shift to higher Pd levels, compared to those at higher temperatures. [2004Bal] pointed to the possibility of the occurrence of a stable or a metastable decagonal (quasicrystalline) phase around the composition  $\text{Al}_{75.5}\text{Fe}_{12.5}\text{Pd}_{12}$  and concluded that further studies are required to clarify this point.

In contrast to the results of [1988Fil] that were reviewed in [1992Rag], [1992Rae] found that, at 500 °C, AlPd and FeAl form a continuous solid solution, and  $\text{Al}_4\text{Pd}$  and  $\text{Al}_3\text{Pd}_2$  dissolve 10 and 23 at.% Fe, respectively.

### Note Added in Proof:

In continuation of their study of the Al-rich alloys of this ternary system, Balanetsky et al. [S. Balanetsky, B. Grushko, T.Ya. Velikanova, and K. Urban, An Investigation of the Al-Pd-Fe Phase Diagram Between 50 and 100 at.% Al: Phase Equilibrium at 750 °C, *J. Alloy. Compd.*, Vol 376, 2004, p 158-164] determined an isothermal section at 750 °C using the same experimental procedures and techniques as above. They found an additional orthorhombic phase denoted N at the composition  $\text{Al}_{76.5}\text{Fe}_{13.0}\text{Pd}_{10.5}$  with approximate lattice parameters of  $a = 2.31$  nm,  $b = 1.60$  nm and  $c = 4.70$  nm. This phase is structurally related to the  $\epsilon$ -phases and decagonal quasicrystals. It is in three-phase equilibrium with O and  $\epsilon$  phases at 750 °C. The C phase was not found at this temperature.

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